

## The Crystal Structure of GaNbO<sub>4</sub>\*

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The crystal structure of gallium niobate has been determined by three-dimensional X-ray methods. Space group *C2* was chosen as the one presenting the most general solution while still retaining the twofold axis established for related structures. The unit-cell dimensions are  $a_0 = 12.660$ ,  $b_0 = 3.7921$ ,  $c_0 = 6.6147$ , and  $\beta = 107.90^\circ$ .

The gallium and niobium atoms occupy distorted octahedral sites similar to those found in titanium niobium oxides and in aluminum niobate.

An alternative centric interpretation is presented at the suggestion of the editor.

### Introduction

The compound GaNbO<sub>4</sub> was prepared at the Sandia Corporation Laboratories during a program of investigation of ferroelectric materials. At the time the compound was prepared, the only Group III niobates or tantalates of the RNbO<sub>4</sub> type for which structural data were available were those of yttrium. Since some question exists as to the structures of these compounds it was hoped that an investigation of the gallium compound might shed some light on the nature of the yttrium compounds or that a new structure type might be found.

Prior to the final refinement of the structure by the present authors, the structure of AlNbO<sub>4</sub> was published by Pedersen (1962). The structure determined by us for GaNbO<sub>4</sub> is substantially in agreement with that reported by Pedersen; the major difference is the assignment of GaNbO<sub>4</sub> to a non-centric space group, whereas Pedersen assigned AlNbO<sub>4</sub> to a centric space group. Pedersen recognized, however, that his two-dimensional data were insufficient to determine the space group firmly.

Some further observations have also been made regarding the nature of the structure of aluminum and gallium niobate and its relationship to other structures, particularly to the titanium niobium oxides described by Wadsley (1961*a, b*).

### Experimental

Crystals of gallium niobate were obtained by flame-fusion of an equimolar mixture of the two oxides (Philgreen, 1963). The fusion was allowed to crystallize on a platinum plate, yielding a radial aggregate of pris-

matic or lath-shaped crystals. The crystals are transparent to translucent, of a very pale greenish tint, 3 mm in length and up to 0.5 mm in diameter.

Equi-inclination Weissenberg photographs were taken with Cu *K* $\alpha$  radiation. These were used for space group determination and the initial determination of the unit-cell dimensions. The final unit-cell dimensions, obtained by means of a modified Picker diffractometer, are as follows:

$$a_0 = 12.660 \pm 0.005 \text{ \AA}$$

$$b_0 = 3.7921 \pm 0.0022$$

$$c_0 = 6.6147 \pm 0.0028$$

$$\beta = 107.90 \pm 0.02^\circ$$

$$(\lambda_{\text{Cu } K\alpha} = 1.5418 \text{ \AA})$$

The diffraction patterns were consistent with monoclinic symmetry and only reflections with  $h+k$  odd were systematically absent. The space group is therefore *C2/m*, *Cm*, or *C2*. The crystals obtained were not of sufficiently good morphological development to aid in the assignment of the space group. A test for piezoelectric effect using the Geibe-Schiebe instrument and a very small sample yielded negative results.

The material available was not suitable for measurement of the specific gravity. With the assumption that this is a true compound rather than a disordered oxide of the RO<sub>2</sub> type in which the metal sites are randomly occupied by Ga and Nb, an even number of empirical formulas per unit cell is required for a *C*-centered cell. The calculated density for four formulas per cell is 5.01 g.cm<sup>-3</sup>, which appears to be reasonable when compared with the density of 5.38 g.cm<sup>-3</sup> for tetragonal YNbO<sub>4</sub> (Barth, 1926). For two and six formulas per unit cell the densities are 2.50 and 7.51 g.cm<sup>-3</sup>, both of which seem unreasonable.

Intensity data for Mo *K* $\alpha$  radiation were collected by means of a General Electric XRD 5 diffractometer and single crystal orienter. Four hundred and twenty-

\* This work was performed under the auspices of the United States Atomic Energy Commission.

two reflections were observed and 76 were unobserved. The crystal used was a rectangular prism of dimensions  $0.015 \times 0.023 \times 0.027$  cm. Lorentz-polarization factors and empirical absorption corrections, obtained by rotation of the crystal about the diffraction vector of several reflections, were applied. Structure factors were calculated using scattering factors for gallium and niobium according to Thomas & Umeda (1957), and for oxygen according to Berghuis, Haanappel, Potters, Loopstra, MacGillavry & Veenendaal (1955). All computations were carried out on a CDC-1604 computer.

### Determination of structure

The corrected intensities were used in computing a three-dimensional Patterson function. Maxima in this function all had  $\nu=0$  or  $\frac{1}{2}$  (or nearly so), indicating that all atoms lay in planes approximately  $b_0/2$  apart. A similar observation ( $h0l$  and  $h2l$  intensity distribution being similar) led Wadsley (1961) to the conclusion that the space group  $A2/m$  was probable for  $\text{TiNb}_2\text{O}_7$ . Pedersen (1962) made the same observation and drew the same conclusion regarding  $\text{AlNbO}_4$ . It should be noted that  $b_0$  for all these compounds is similar, about  $3.8 \text{ \AA}$ . Since no positive evidence is available for establishing the space group as centric or non-centric, a non-centric space group was chosen as representing the more general case. If the structure is in fact centric, this should become apparent in that the atomic parameters should approach the centric positions upon refinement. In view of the establishment of the twofold axes in  $\text{TiNb}_2\text{O}_7$  and  $\text{AlNbO}_4$ , the space group  $C2$  was chosen in preference to  $Cm$ .

The absence of Patterson function maxima of the type  $0, \nu, \frac{1}{2}$  or  $\frac{1}{2}, \nu, 0$  eliminates the twofold positions of  $C2$  (or  $C2/m$ ), and therefore the Ga and Nb must be in general positions if  $C2$  is assumed. With this assumption the approximate parameters of Ga and Nb were determined from the Patterson function.

Oxygen parameters were obtained from a Fourier synthesis using phases calculated from the Ga and Nb parameters determined above. Refinement by several difference and differential syntheses (shifts  $\leq 0.1\sigma$ ) yielded the parameters listed in Table 1 ( $y$  of niobium arbitrarily set to 0.0) with the estimated standard deviations in positional parameters calculated by the method of Cruickshank (1949). The final tabulation for observed and calculated structure factors is given in Table 2. The final  $R$  value was 0.080 using only

observed and unextinguished reflections. The value increases to 0.092 using all observed and unobserved reflections.

### Discussion of the structure

An examination of the final parameters shows that all atoms have a  $y$  parameter close to zero. The  $y$  parameters of the gallium and two oxygen atoms, however, are more than ten standard deviations from zero. On this basis we conclude that the correct space group for  $\text{GaNbO}_4$  is  $C2$ . It is likely that this is also true for  $\text{AlNbO}_4$  and possibly for  $\text{TiNb}_2\text{O}_7$ . Both Pedersen and Wadsley consider this possibility, but defer a final conclusion because of their limited (two-dimensional) data. The low value of the thermal parameters,  $B$ , shown in Table 1 are consistent with those obtained in high refractory materials and with those observed for  $\text{BaTiO}_3$  (Evans, 1961) and  $\text{NaNbO}_4$  (Wood, Miller & Remeika, 1962).

Fig. 1 illustrates the structure of  $\text{GaNbO}_4$  as viewed along  $b_0$ . The illustration is essentially that of Pedersen except that bonds rather than coordination groups are shown. The coordination about gallium and niobium is a highly distorted octahedral one. The bond lengths and bond angles are listed in Table 3. The distortion of the octahedra is clearly related to the sharing of octahedral edges. The O-O distance along shared edges varies from  $2.50$  to  $2.70 \text{ \AA}$ , whereas the

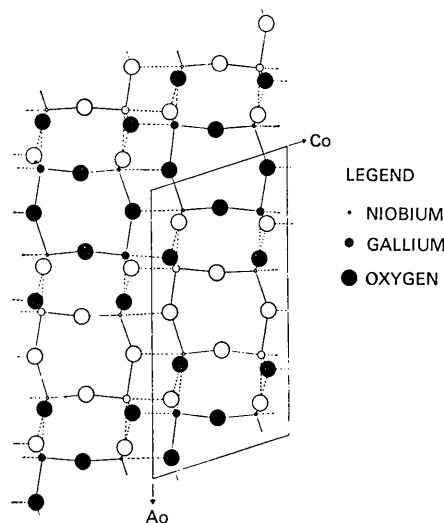


Fig. 1. The structure  $\text{GaNbO}_4$  viewed down the  $b_0$  axis. Filled circles indicate atoms situated near  $y=0.0$ . Open circles indicate atoms situated near  $y=0.5$ .

Table 1. Atomic positions and temperature factors

	$x$	$10^{-5}\sigma(x)$	$y$	$10^{-5}\sigma(y)$	$z$	$10^{-5}\sigma(z)$	$B(\text{\AA}^2)$
Ga	0.19711	7	0.00768	32	0.81355	16	0.073
Nb	0.10317	5	0.00000	22	0.23054	12	0.299
O(1)	0.25943	45	-0.00635	219	0.14614	108	0.300
O(2)	0.36246	46	0.00868	227	0.79903	112	0.402
O(3)	0.05483	43	0.02533	211	0.86090	104	0.174
O(4)	0.13734	46	-0.02390	226	0.51466	111	0.391



O–O distance for adjacent oxygens on unshared edges varies from 2.73–3.26 Å. Although the average bond angle involving two adjacent oxygens and either metal is very close to the undistorted octahedral value of 90°, the range is from 72.7° to 108.6°.

It is interesting to note that for both gallium and niobium the four shortest metal–oxygen bonds (first four distances in Table 3) form a distorted tetrahedron. For both metals the average bond angle (first six bond angles in Table 3) for this distorted tetrahedral group

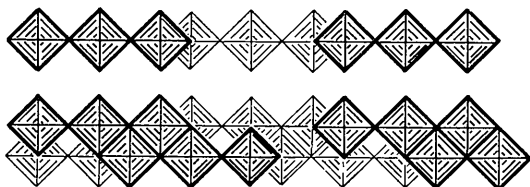


Fig. 2. Upper part of the figure illustrates a unit of the hypothetical  $R_3O_8$  structure formed by fusing together the blocks of octahedra. Each hatched square represents an octahedron in projection as viewed down a corner; the lighter lines indicate blocks displaced downwards by half an octahedral diagonal. Two of these  $R_3O_8$  units can be joined together (by translating the lower unit sideways by  $1\frac{1}{2}$  octahedral diagonals) to form the double octahedral sheet.

is within a few degrees of the tetrahedral angle, but one of the six angles in each case is very large (about 149°). In Fig. 1, the longer metal–oxygen bonds are shown by broken lines. Tetrahedrally coordinated niobium is reported by Komkov (1959) in fergusonite (natural  $YNbO_4$ ). The crystals display an excellent cleavage parallel to (001) and a good cleavage parallel to (100). Both cleavages involve rupturing of the longer, and thus weaker, metal–oxygen bonds.

Another interesting feature of the structure is the rather large void centered at  $x=0$ ,  $y=\frac{1}{2}$ ,  $z=\frac{1}{2}$ . The shortest distance between this point and a neighboring oxygen position is 2.62 Å. An ion with a radius somewhat greater than 1 Å could be easily accommodated in this void. No structures analogous to that of  $GaNbO_4$  with the void occupied are known to the authors.

Wadsley (1961*b*) proposes a structural scheme for a series of disordered titanium niobium oxides of composition  $R_{3n}O_{8n-3}$ . This series includes both known and hypothetical members. The structures are formed from units consisting of three octahedra forming a short chain by sharing of corners. These units are linked into 'single' or 'double' layers in the manner shown in Fig. 2 by edge and corner sharing. The several mem-

Table 3. *Interatomic separations and angles*

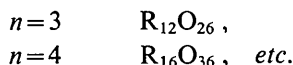
Ga–O(4a)	1.891 Å	Nb–O(4a)	1.797 Å
Ga–O(3a)	1.918	Nb–O(3e)	1.906
Ga–O(1b)	1.922	Nb–O(2c)	1.937
Ga–O(1c)	2.021	Nb–O(2b)	2.000
Ga–O(1d)	2.097	Nb–O(1a)	2.215
Ga–O(2a)	2.125	Nb–O(3f)	2.333
O(2a)–O(1c)	2.498	O(3e)–O(3f)	2.631
O(1a)–O(2c)	2.498	O(1d)–O(1b)	2.666
O(2a)–O(1b)	2.583	O(1d)–O(1c)	2.666
O(1a)–O(2b)	2.583	O(3a)–O(1d)	2.698
O(4a)–O(2c)	2.730	O(3e)–O(2c)	2.934
O(3a)–O(4a)	2.800	O(3a)–O(1c)	2.959
O(3f)–O(2b)	2.840	O(4a)–O(1b)	2.965
O(4a)–O(1c)	2.879	O(2a)–O(1d)	2.969
O(4a)–O(3e)	2.894	O(3e)–O(2b)	3.096
O(2a)–O(4a)	2.895	O(3a)–O(1c)	3.109
O(4a)–O(2b)	2.896	O(1a)–O(4a)	3.264
O(3f)–O(2c)	2.924		
O(3a)–Ga–O(4a)	94.51°	O(4a)–Nb–O(2c)	93.89°
O(4a)–Ga–O(1c)	94.71	O(4a)–Nb–O(2b)	99.28
O(3a)–Ga–O(1b)	100.78	O(3e)–Nb–O(2b)	99.55
O(4a)–Ga–O(1b)	102.22	O(4a)–Nb–O(3e)	102.78
O(3a)–Ga–O(1c)	104.06	O(3e)–Nb–O(2c)	104.85
O(1c)–Ga–O(1b)	148.60	O(2b)–Nb–O(2c)	148.80
O(2a)–Ga–O(1b)	76.12	O(1a)–Nb–O(3f)	72.73
O(2a)–Ga–O(1c)	77.03	O(1a)–Nb–O(2b)	73.64
O(1d)–Ga–O(1c)	80.66	O(1a)–Nb–O(2c)	75.39
O(1d)–Ga–O(1b)	83.06	O(3e)–Nb–O(3f)	75.98
O(3a)–Ga–O(1d)	84.21	O(3f)–Nb–O(2c)	81.53
O(2a)–Ga–O(1d)	89.38	O(3f)–Nb–O(2b)	85.90
O(2a)–Ga–O(4a)	92.07	O(4a)–Nb–O(1a)	108.46

In parenthesis for each oxygen atom are letters which refer to the following symmetry operations:

<i>a</i>	$x, y, z$
<i>b</i>	$\frac{1}{2}-x, \frac{1}{2}+y, 1-z$
<i>c</i>	$\frac{1}{2}-x, -\frac{1}{2}+y, 1-z$
<i>d</i>	$x, y, 1+z$
<i>e</i>	$-x, -y, 1-z$
<i>f</i>	$x, y, -1+z$

bers of the series may now be formed by corner-sharing between successive single and double layers in the appropriate ratio. A structure consisting of purely single layers has the composition R<sub>3</sub>O<sub>8</sub> (not a part of the series).

The composition, GaNbO<sub>4</sub> or R<sub>2</sub>O<sub>4</sub>, does not appear in Wadsley's proposed series. The configuration of shared octahedra in GaNbO<sub>4</sub>, however, bears a strong resemblance to part of Wadsley's double layers. A structural scheme is here proposed which will include a compound of the R<sub>2</sub>O<sub>4</sub> type with a structure corresponding to that of GaNbO<sub>4</sub>. This scheme differs from Wadsley's in that the fundamental unit consists of two rather than three octahedra. The series of composition R<sub>4n</sub>O<sub>10n-4</sub> is formed, then, by assembling the units into single and double layers and linking these in appropriate sequence as shown in Fig. 3. A structure consisting of single layers alone would have the composition R<sub>2</sub>O<sub>5</sub>, again not fitting into the series. If  $n=2$ , we have R<sub>8</sub>O<sub>16</sub> consisting only of double layers. If we consider this to be an ordered structure involving two metals we can write Ga<sub>4</sub>Nb<sub>4</sub>O<sub>16</sub>, the unit-cell contents for the present compound. Other members of the series would be:



The compound R<sub>4</sub>O<sub>6</sub>, for  $n=1$ , cannot be made by any combination of double and single layers. No structures of this type corresponding to these compounds are known to us.

Wadsley proposes both a monoclinic and orthorhombic structure in his series with  $n=6$ . This compound has the composition R<sub>2</sub>O<sub>5</sub>. In the scheme proposed here the single-layer structure has the composition R<sub>2</sub>O<sub>5</sub> and would be orthorhombic, C-centered, with dimensions near  $a_0=12.4$ ,  $b_0=3.7$ ,  $c_0=4.2$  Å. No detailed structures for Nb<sub>2</sub>O<sub>5</sub> or Ta<sub>2</sub>O<sub>5</sub> have been reported, but several reported cell dimensions are related to these values or to simple multiples or fractions thereof (Lapitskii, Simanov, Semenenko & Yarembash, 1954; Simanov, Lapitskii & Artamonova, 1954).

The assistance of Mr H. Bezdek in collecting the intensity data, the useful discussions with Dr J. R. Brathovde (National Science Foundation), the computer assistance of Dr F. H. Kruse (Los Alamos

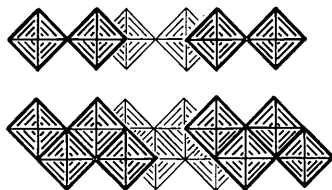


Fig. 3. The hypothetical R<sub>2</sub>O<sub>4</sub> structure is formed with two octahedra compared with three for R<sub>3</sub>O<sub>8</sub> (Fig. 2). Two of these R<sub>2</sub>O<sub>4</sub> units (top of figure) can be joined together by translation of the lower unit sideways by  $\frac{1}{2}$  octahedral diagonals to form the double layer R<sub>8</sub>O<sub>16</sub> (lower figure) of which GaNbO<sub>4</sub> is a member.

Scientific Laboratory), and the piezoelectric examination by Dr Robert Douglas (Los Alamos Scientific Laboratory), are acknowledged.

## APPENDIX

The referee has objected to our choice of the noncentric space group, especially since the difference of the  $y$  coordinates from zero between a centric and noncentric space group would probably be strongly correlated with thermal displacements parallel to  $b_0$  (*i.e.*, the  $B_{22}$  term using anisotropic thermal parameters). We have refined, with several cycles of differential synthesis, the positional and anisotropic thermal parameters for a centric model. The final  $R$  value of 0.078 using only observed reflections (0.092 for all reflections) was not significantly different from that obtained with the noncentric case. These centric parameters are presented in Table 4\*.

Table 4. Atomic positions and anisotropic temperature factors for centric interpretation

	$x$	$z$	$B_{11}$	$B_{22}$	$B_{33}$	$B_{13}$
Ga	0.19712	0.81359	0.084	0.303	0.044	0.082
Nb	0.10320	0.23062	0.302	0.449	0.237	0.058
O(1)	0.25941	0.14594	0.312	0.331	0.294	-0.012
O(2)	0.36252	0.79879	0.487	0.082	0.295	-0.047
O(3)	0.05492	0.86129	0.164	0.799	0.325	0.190
O(4)	0.13705	0.51397	0.567	1.319	0.104	0.011

The editor has pointed out that a  $y$  coordinate of 0.025, representing a displacement of the oxygen atoms from the symmetry plane by about 0.1 Å, is essentially indistinguishable from a  $y$  coordinate of zero and a  $B_{22}$  term of such a magnitude as to give an extra 0.1 Å r.m.s. thermal displacement along  $b$ . This appears to be the case for oxygen atoms O(3) and O(4). The average of the difference between the r.m.s. displacements along  $b_0$  and the next largest principal axis for these atoms is 0.06 Å. This perhaps would be predictable since these oxygen atoms do not have any constraining bonds directed along  $b_0$  whereas O(1) and O(2) do. In addition the anisotropic thermal parameters for O(1) and O(2) are in general agreement with their respective regular and irregular arrangement of nearest neighbors. Similarly the differences of the r.m.s. displacement along  $b_0$  and the next largest principal axis for the Ga atom corresponds very closely to the 0.03 Å displacement from the symmetry plane obtained in the noncentric refinement.

A satisfactory solution for our intensity data may be obtained either with the centric space group with anisotropic refinement of thermal parameters or with the noncentric space group and isotropic thermal para-

\* A table of observed and calculated structure factors has not been included for the centric refinement. Values are not significantly different from those listed in Table 2. Interatomic separations are essentially the same as Table 3 with approximate averages to be taken between atoms related by the symmetry operations ( $b$ ) and ( $c$ ).

meters. We believe that where there are two solutions possible, it is preferable to choose the more general case, *i.e.* the noncentric interpretation.

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## The Crystal Structure of $\text{NaCl} \cdot 5\frac{1}{7}\text{NH}_3$ and $\text{NaBr} \cdot 5\frac{1}{7}\text{NH}_3$

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The crystal structures of  $\text{NaCl} \cdot 5\frac{1}{7}\text{NH}_3$  and  $\text{NaBr} \cdot 5\frac{1}{7}\text{NH}_3$  have been determined from single-crystal X-ray data, collected in a low-temperature Weissenberg camera with Cu K radiation. Both compounds are trigonal, space group  $R\bar{3}$ , with 21 formula units in a hexagonal cell for which the measured dimensions are:

$$\begin{aligned} \text{NaCl} \cdot 5\frac{1}{7}\text{NH}_3: a &= 23.67, c = 9.25 \text{ \AA} \quad (t = -58^\circ \text{C}); \\ \text{NaBr} \cdot 5\frac{1}{7}\text{NH}_3: a &= 24.34, c = 9.46 \text{ \AA} \quad (t = -35^\circ \text{C}). \end{aligned}$$

The ammonia molecules and halide ions together form a roughly cubic close-packed arrangement, the sodium ions occupying certain of the octahedral interstices, so that only ammonia molecules are closest neighbours. The octahedra, which enclose some of the sodium ions, share two corners with adjacent octahedra; in this case the effective coordination of the sodium is only fivefold, the sixth ammonia molecule lying at a greater distance.

### Introduction

The present investigation forms part of a series of structure determinations of some amines formed with salts of the alkali metals and ammonium. The structures of  $\text{NH}_4 \cdot 3\text{NH}_3$  (X=Cl, Br, I) and  $\text{NH}_4\text{I} \cdot 4\text{NH}_3$  have been reported earlier (Olovsson, 1960 *a, b*).

In the systems  $\text{NaX} \cdot \text{NH}_3$  (X=F, Cl, Br, I) the following compounds have been reported:  $\text{NaCl} \cdot 2.5\text{NH}_3$  (Friedrichs, 1921),  $\text{NaCl} \cdot 5\text{NH}_3$  (Joannis, 1891; Biltz & Hansen, 1923; Biltz, 1923; Patscheke, 1933),  $\text{NaBr} \cdot 5.25\text{NH}_3$  and  $\text{NaBr} \cdot 5.75\text{NH}_3$  (Biltz & Hansen, 1923; Biltz, 1923),  $\text{NaI} \cdot 4\text{NH}_3$  (Picon, 1919; Leonard, Lippincott, Nelson & Sellers, 1955),  $\text{NaI} \cdot 4.5\text{NH}_3$  and  $\text{NaI} \cdot 6\text{NH}_3$  (Biltz & Hansen, 1923; Biltz, 1923). A phase with the composition  $\text{NaCl} \cdot 2.5\text{NH}_3$  does not exist according to later investigations (Biltz & Hansen, 1923). The phase reported as  $\text{NaI} \cdot 4\text{NH}_3$  has the composition  $\text{NaI} \cdot 4.5\text{NH}_3$  according to Biltz & Hansen (1923). The structures of the amines of sodium

iodide are currently being investigated in this laboratory.

In this paper the structures of the compounds formulated above as  $\text{NaCl} \cdot 5\text{NH}_3$  and  $\text{NaBr} \cdot 5.25\text{NH}_3$  are reported. The X-ray crystallographic investigation shows that these compounds are isomorphous and have the exact composition  $\text{NaCl} \cdot 5\frac{1}{7}\text{NH}_3$  and  $\text{NaBr} \cdot 5\frac{1}{7}\text{NH}_3$ , respectively. This point is further discussed below. For convenience these compounds are called 'pentammines' in the following text, although it should be kept in mind that this designation is not altogether adequate.

### Experimental

The solubility curve of sodium chloride in ammonia is shown in Fig. 1. The values were taken from the work by Patscheke (1933) (*cf.* also Patscheke & Tanne, 1935), but the results of Guyer, Bieler & Schmid (1934) and Distanov (1938) are very similar. The curves give the composition of the solution in equili-