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The Crystal Structure of GaNbO₄*

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The crystal structure of gallium niobate has been determined by three-dimensional X-ray methods. Space group C2 was chosen as the one presenting the most general solution while still retaining the twofold axis established for related structures. The unit-cell dimensions are $a_0 = 12.660$, $b_0 = 3.7921$, $c_0 = 6.6147$, and $\beta = 107.90^\circ$.

The gallium and niobium atoms occupy distorted octahedral sites similar to those found in titanium niobium oxides and in aluminum niobate.

An alternative centric interpretation is presented at the suggestion of the editor.

Introduction

The compound GaNbO₄ was prepared at the Sandia Corporation Laboratories during a program of investigation of ferroelectric materials. At the time the compound was prepared, the only Group III niobates or tantalates of the RNbO₄ type for which structural data were available were those of yttrium. Since some question exists as to the structures of these compounds it was hoped that an investigation of the gallium compound might shed some light on the nature of the yttrium compounds or that a new structure type might be found.

Prior to the final refinement of the structure by the present authors, the structure of $AlNbO_4$ was published by Pedersen (1962). The structure determined by us for GaNbO₄ is substantially in agreement with that reported by Pedersen; the major difference is the assignment of GaNbO₄ to a non-centric space group, whereas Pedersen assigned AlNbO₄ to a centric space group. Pedersen recognized, however, that his two-dimensional data were insufficent to determine the space group firmly.

Some further observations have also been made regarding the nature of the structure of aluminum and gallium niobate and its relationship to other structures, particularly to the titanium niobium oxides described by Wadsley (1961a, b).

Experimental

Crystals of gallium niobate were obtained by flamefusion of an equimolar mixture of the two oxides (Philgreen, 1963). The fusion was allowed to crystallize on a platinum plate, yielding a radial aggregate of prismatic or lath-shaped crystals. The crystals are transparent to translucent, of a very pale greenish tint, 3 mm in length and up to 0.5 mm in diameter.

Equi-inclination Weissenberg photographs were taken with Cu $K\alpha$ radiation. These were used for space group determination and the initial determination of the unit-cell dimensions. The final unit-cell dimensions, obtained by means of a modified Picker diffractometer, are as follows:

$$a_0 = 12.660 \pm 0.005 \text{ A}$$

$$b_0 = 3.7921 \pm 0.0022$$

$$c_0 = 6.6147 \pm 0.0028$$

$$\beta = 107.90 \pm 0.02^{\circ}.$$

$$(\lambda \text{Cu } K\alpha = 1.5418 \text{ Å})$$

The diffraction patterns were consistent with monoclinic symmetry and only reflections with h+k odd were systematically absent. The space group is therefore C2/m, Cm, or C2. The crystals obtained were not of sufficiently good morphological development to aid in the assignment of the space group. A test for piezoelectric effect using the Geibe-Schiebe instrument and a very small sample yielded negative results.

The material available was not suitable for measurement of the specific gravity. With the assumption that this is a true compound rather than a disordered oxide of the RO₂ type in which the metal sites are randomly occupied by Ga and Nb, an even number of empirical formulas per unit cell is required for a *C*-centered cell. The calculated density for four formulas per cell is 5.01 g.cm^{-3} , which appears to be reasonable when compared with the density of 5.38 g.cm^{-3} for tetragonal YNbO₄ (Barth, 1926). For two and six formulas per unit cell the densities are $2.50 \text{ and } 7.51 \text{ g.cm}^{-3}$, both of which seem unreasonable.

Intensity data for Mo $K\alpha$ radiation were collected by means of a General Electric XRD 5 diffractometer and single crystal orienter. Four hundred and twenty-

^{*} This work was performed under the auspices of the United States Atomic Energy Commission.

two reflections were observed and 76 were unobserved. The crystal used was a rectangular prism of dimensions $0.015 \times 0.023 \times 0.027$ cm. Lorentz-polarization factors and empirical absorption corrections, obtained by rotation of the crystal about the diffraction vector of several reflections, were applied. Structure factors were calculated using scattering factors for gallium and niobium according to Thomas & Umeda (1957), and for oxygen according to Berghuis, Haanappel, Potters, Loopstra, MacGillavry & Veenendaal (1955). All computations were carried out on a CDC-1604 computer.

Determination of structure

The corrected intensities were used in computing a three-dimensional Patterson function. Maxima in this function all had v=0 or $\frac{1}{2}$ (or nearly so), indicating that all atoms lay in planes approximately $b_0/2$ apart. A similar observation (h0l and h2l intensity distribution being similar) led Wadsley (1961) to the conclusion that the space group A2/m was probable for TiNb₂O₇. Pedersen (1962) made the same observation and drew the same conclusion regarding AlNbO₄. It should be noted that b_0 for all these compounds is similar, about 3.8 Å. Since no positive evidence is available for establishing the space group as centric or non-centric, a non-centric space group was chosen as representing the more general case. If the structure is in fact centric, this should become apparent in that the atomic parameters should approach the centric positions upon refinement. In view of the establishment of the twofold axes in $TiNb_2O_7$ and $AlNbO_4$, the space group C2 was chosen in preference to Cm.

The absence of Patterson function maxima of the type 0, $v, \frac{1}{2}$ or $\frac{1}{2}$, v, 0 eliminates the twofold positions of C2 (or C2/m), and therefore the Ga and Nb must be in general positions if C2 is assumed. With this assumption the approximate parameters of Ga and Nb were determined from the Patterson function.

Oxygen parameters were obtained from a Fourier synthesis using phases calculated from the Ga and Nb parameters determined above. Refinement by several difference and differential syntheses (shifts $\leq 0.1\sigma$) yielded the parameters listed in Table 1 (y of niobium arbitrarily set to 0.0) with the estimated standard deviations in positional parameters calculated by the method of Cruickshank (1949). The final tabulation for observed and calculated structure factors is given in Table 2. The final R value was 0.080 using only observed and unextinguished reflections. The value increases to 0.092 using all observed and unobserved reflections.

Discussion of the structure

An examination of the final parameters shows that all atoms have a y parameter close to zero. The y parameters of the gallium and two oxygen atoms, however, are more than ten standard deviations from zero. On this basis we conclude that the correct space group for GaNbO₄ is C2. It is likely that this is also true for AlNbO₄ and possibly for TiNb₂O₇. Both Pedersen and Wadsley consider this possibility, but defer a final conclusion because of their limited (two-dimensional) data. The low value of the thermal parameters, *B*, shown in Table 1 are consistent with those obtained in high refractory materials and with those observed for BaTiO₃ (Evans, 1961) and NaNbO₄ (Wood, Miller & Remeika, 1962).

Fig. 1 illustrates the structure of GaNbO₄ as viewed along b_0 . The illustration is essentially that of Pedersen except that bonds rather than coordination groups are shown. The coordination about gallium and niobium is a highly distorted octahedral one. The bond lengths and bond angles are listed in Table 3. The distortion of the octahedra is clearly related to the sharing of octahedral edges. The O-O distance along shared edges varies from 2.50 to 2.70 Å, whereas the



Fig. 1. The structure GaNbO₄ viewed down the b_0 axis. Filled circles indicate atoms situated near y=0.0. Open circles indicate atoms situated near y=0.5.

Table 1. Atomic positions and temperature factors

	x	$10^{-5}\sigma(x)$	у	$10^{-5}\sigma(y)$	z	$10^{-5}\sigma(z)$	B(Å2)
Ga	0.19711	7	0.00768	32	0.81355	16	0.073
Nb	0.10317	5	0.00000	22	0.23054	12	0.299
O(1)	0.25943	45	-0.00635	219	0.14614	108	0.300
O(2)	0.36246	46	0.00868	227	0.79903	112	0.402
O(3)	0.05483	43	0.02533	211	0.86090	104	0.174
O(4)	0.13734	46	-0.02390	226	0.51466	111	0.391

Table 2. Observed and calculated structure factors for GaNbO4

The plus sign before the observed values should be interpreted as 'less than'.

н	10F0 10FC 104	108	н 10	FO 10FC 10A	10B	н	10F0	10FC 10/	10B	н	10	F0 10	FC 10A	108	н	10F0	10FC	104	10B
N 4 6 0	K=0 L=0 666 630 -630 926 1094-1094 556 601 -601	0 0 0	K 1 + 3 7 5 2 7 13	43 L1 96 92 80 46 845 845 64 229 229 46 1550-1548	-46 -33 -9	7 9 11 13	359 389 +124 1063	367 -36 365 -36 71 -6 1067 106	3 48 5 12 5 -26 5 25	15	6 6	58 6 40 6	00 -599 55 655 4	-23 10	10 12 14 16	+115 368 943 +140	24 304 - 902 48	-24 304 902 -48	0 0 0
10 12 14 16	1140 1173 1173 447 411 -411 574 522 -522 +136 36 36	0 1 0 1 - 0	9 2 1 3 3 5 K	48 197 193 108 313 305 181 550 549	42 -69 28	0246	K-2 1774 1434 836 730	L-3 1492-149 1321 132 856 854 705 -70	1 -66 1 5 5 8 5 -14	6 8 10	156 68 40 51 +1	51 17 39 5 66 4 92 5 21	56-1756 94 588 60 460 13 513 89 52	-16 81 -17 72	1 3 5 7	K=1 461 702 1061 898	L= -6 386 - 682 1037 1 853 -	385 682 035 853	-14 9 53 -38
1 3 5 7	K 1 L 0 1513 1700 1700 955 1149-1149 269 390 -386 741 800 -797	21 18 54 -75 1	2 2 4 7 6 8 4 4	26 274 265 94 863 863 39 1070-1066 73 479 -479 35 422 421	-71 12 -98 4 27	10 12	261 304 667 K 3	266 -244 308 -30 658 654 L 3 289 -28	5 -41 5 -41 3 14	. 14 16	10 +1	10 9 40 1 =3 1 49 8	.64 164 4 .81 881 863	-68 5 26	11 13 15	413 158 364 794	423 - 150 - 315 757	423 150 315 757	-9 -11 0
11 13 15	835 811 810 975 951 -951 +132 51 -49	18 -27 13	і ні з 5 5 Н	35 L -1 35 84 57 31 566 566 38 151 151	-61 -23 1	3579	1489 254 305 327	1363 136 239 -23 317 -29 309 •30	3 30 -48 3 106 3]]	11	7 7 3	99 7 57 7 50 3	99 -799 13 710 133 329 170 560	27 56 - 45 109	2468	199 1150 166 689	109 - 1212 1 166 - 705 -	106 209 166 704	-26 84 -38
0246	K-2 L-0 2151 2714 2712 480 447 -447 711 821 -820 428 483 -483	91 -2 42	K 0 20 2 5	0 2 2 11 2116-2116 02 332 332	0	11 0	+140 K 4 865 810	94 -6	7 -66	13	к, +12	92 7	42 -741 58 157	-34 7	10 12 14	+124 309 887	36 278 - 834 -	-14 272 834.	33 -57 5
8 10 12 14	+100 111 -15 1049 1076 1075 408 368 -366 525 482 -482	-110 45 -43 -1 1	0 15 8 + 0 12 2 2 4 2	53 1491 1491 97 59 -59 38 1274-1274 36 231 231 00 165 165	0 0 0 0	6	543 502 K 0 391	606 60 535 -53 L -3 491 -491	5 13 5 -29	6 8 10	44 35 43	56 3 33 3	16 400 19 318 85 385	115 -27 4	1 3 5 7 9	373 565 849 752 361	329 -: 543 857 8 727 -: 339 -	327 543 848 721 338	-34 18 125 -92 20
1 3 5 7	K 3 L 0 803 1039 1038 740 801 -801 191 279 -256 607 615 -597	39 -1 109 -147	K 1 5 3 4 2	1 L 2 03 589 -588 24 362 379 28 310 310	28 44 - 1	6 8 10 12	702 996 907 951 •108	638 -638 1065 1069 888 888 895 -899 33 -33	5 0 5 0 5 0 5 0	0 2 4 6 8	12 12 +1 +1	33 11 56 12 59 4 15 18 3	11 1111 38-1238 69 -469 0 0 91 391	0 0 0 0	11 2 4	+137 K: 4 +140- 876	120 - L -6 116 - 942	118 107 933	-24 -45 126
11 13	623 623 622 694 677 676 733 803 -800 K 4 L 0	-24 43 -70 I	13 9 11 1 5 3 2	79 1429 1429 00 1120-1120 81 565 -565 77 292 292	30 -13 -5	14	258 418 K 1 879	241 -241 370 370 L3 1018-1016		10 1 3	70 K- +10 116	-1 L 52 12	24 724 -5 -144 44-1144	0 - 17 - 36	6 0 2	+141 K+0 847 830	131 - 691 - 733	130 691 733	-9 0
2 # 6 8 0	262 283 -281 *88 526 -524 317 328 -327 -129 154 -38	-24 46 19 -149		21 1641-1637 80 275 258 91 128 125 78 1239 1239	-116 97 28 33	357911	483 433 1411 97 849	292 -28 1595 159 42 -42 826 -826	-18 -73 66 5	5 7 9	21 25 89 K	40 2 56 2 97 9	48 -247 08 908	-19 9	6	358 264 K=1 419	300 255 L •7 380	300 255 378	0 0 34
1 3 5	K 5 L 0 559 663 662 474 550 -550 165 198 -159	-22 124	0 11 2 2 K	30 1147-1146 02 224 218 3 L 2 63 450 4457	-30	15	123 145 974 K 2	158 -156 952 952 L3	-7	2 4 6 8	111	25 4 26 3	02-1098 02-1098 12 -412 9 7 57 357	-94 -17 5 -16	35	624 +135 K-2 723	542 104 L=7 631 -	542 103 631	-13 -12
n 2	K=0 L 1 749 787 787 990 1108-1108 996 1072-1072	0 0 10	3 2 2 10 9 9	77 276 266 06 250 250 47 1079 1076 32 919 -919 66 457 -457	74 -17 76 -36	6 8 10	5*9 856 819 867 +118	544 -537 881 881 804 800 819 -816 74 -42	-83 -2 83 -42 61	1 3 5 7	K+ +12 93 18	-3 L 21 1 31 9 36 1	-5 11 -105 38 -934 78 175 97 -201	- 38 -83 35 -50	1	327 K 3 317	274 2 L=7 329	317	-51 87
6 5 10 12 13	1553 1564 1564 162 199 199 -103 31 -31 504 491 -491 910 793 -793	0 0 0 0	K 0 13 2 20 4 13	4 L.2 12 1110-1102 - 09 220 181 24 71 70	-132 126 9	14 16 1	239 390 K-3 633	219 -219 344 344 L3 709 -709	-2	0	к 71 82	4 L 39 7 24 8	66 757 47 -836	113 -139	2 4 6 8	K+0 887 981 +115 543	L= -7 783 - 929 20 477	783 929 20 477	0000
16 1 3	788 772 772 K=1 L 1 440 343 322 1067 1248-1248	0 118 - 30	68 51 18 12	12 871 870 36 45 -41 5 L 2 68 317 -310	47 16 63	3 5 7 9 11	304 279 1190 119 715	357 - 357 257 -213 1250 1242 65 -65 686 -686	-14 -143 137 4 -7	2 4 6 8	K- 54 11 11	0 L 9 5 3 1 12 1 33 10	5 87 587 20 -120 65 -165 54-1054	0 0 0	10 12 14	794 678 618 K 1	739 - 617 - 575 - L= -7	739 617 575	000
5 7 9 11 13	130 103 101 1071 1139 1139 226 159 -156 211 166 166 1197 1201-1201	22 23 -21 0 -25	5 1 K 2 1 4 14	40 199 184 0 L -2 08 96 96 34 1432 1432	0 0	13	+134 K 4 201 405 635	80 56 L= -3 231 -226 409 -395	5 57 5 - 102	10 12 14 16	119 39 39	59 11 90 3 91 3 77 3	06 1106 80 380 48 -348 32 -332	0 0 0 0	13579	1357 262 553 347 +123	1227-12 202 520 330 19	227 201 520 336 9	-2 20 -14 -8 16
024	K 2 L 1 481 575 563 700 847 -845 906 941 -939	114 14 48 14 -61 11	8 2 0 11 2 1 4 10 6 +1	51 136 136 07 1137-1137 90 151 151 90 1086 1086 32 117 -117	0 0 0 0	810	623 662 × 5 423	628 618 652 -649 L3 466 -466	113 -68	1 3 5 7	107 11 22 102	13 10 18 1 23 1 27 10	39 1038 71 168 37 131 20-1020 29 -129	37 -28 39 5 7	13	238 K=2 797 877	236 2 L7 713 -7	236 713	24
6 8 10 12 14	1218 1303 1298 172 196 192 +114 41 -20 476 434 -434 750 728 -727	113 -39 -35 -7 -34	K 1 12 3 8 5 4	1 L · -2 98 1379-1376 01 867 866 18 505 503	-92 32 -44	3 0 2	237 K 0 897 254	227 -227 L-4 824 824 274 274	-2	11 13 15	136 4 26 X	2 13 33 3 37 2 2 L	84 1384 71 - 369 70 - 269 = -5	- 36 11	6 8 10 12	+125 484 737 616	64 446 674 -6 573 -1	19 445 673 570	-60 28 -19 -55
1357	K 3 L 1 264 281 207 897 1032-1030 +105 100 91	190 1 -67 1 *3 1	9 3 1 11 3 12 5 4	83 24 -15 51 393 -393 57 1166-1166 47 1259 1259 38 414 414	-18 10 -3 37 -7	4 8 10 12	• 96 1261 439 803 •136	106 -106 1296-1296 391 391 788 788 61 -61	0 0 0 0	2 4 6 8 10	49 13 14 96	1 5 38 1 15 1 16 9 10 9	11 511 20 -118 82 -166 39 -939 94 994	-11 -18 73 -1 27	1 3 5 7	K=3 1116 225 461 311	L= -7 1034-10 177 426 280	034 169 424 279	-11 53 -36 -21
9 11 13	144 138 -121 +130 138 138 1026 1010-1008	-67 5 -65	7 2. K 2 4 4 10	57 237 -237 2 Lz2 94 76 73 36 1197 1196	-2 -20 44	1357	K-1 + 89 249 599	L-4 153 136 192 191 593 -593	-70	12	38 34 86	10 3 12 3 13 L	48 945 17 -316 5 72 868	-46 -24 84	9	+139 K=0 +130 1097	50 L=8 53 994	26 -53 994	43 0
0 2 # 6 8	299 372 354 470 564 -561 657 722 -718 883 940 927 +132 166 154	114 54 14 -83 14 150 1	6 1 0 10 2 1 4 10	00 350 -330 - 74 104 103 12 1009-1009 43 136 132 09 987 987	-110 19 7 33 34	9 11	881 +132 K 2 729	971 971 128 128 L=4	32	3 5 7 9 11	+12	1 1 6 8 6 3 11	24 82 21 -821 99 -98 39 1139	-70 92 6 13 25	4 1 3	+141 K=1 553 653	1.•8 446 4 693 (-87 446 593	6 29
10 1 3	+139 48 -8 K-5 L-1 +135 242 146 608 761 -757	- 47 193 -79	x 1 10 3 73 5 3	3 L+ -2 87 1015-1003 - 21 773 770 29 393 383	-157 73 -85	2468 10	237 +107 1064 358 727	244 239 85 -60 1172-1170 350 347 723 721	-49 60 -72 39 44	2	X- 38 +13 18	4 L 4 3 3 1	5 74 374 10 -107 74 -137	-22 -25 107	0	K=2 +138 K=0 +127	L=8 74 L= -8 66	-56	-48 0
2 4	+141 81 64 K-0 L1 483 579 579 1633 1703 1703	49 0 0 0	7 +1 9 3 1 9 3 10	09 39 -24 28 321 -320 64 934 -934 45 1043 1038	-31 28 -17 92	1 3 5	K-3 +113 201 478	L-4 194 126 209 202 491 -491	-147 51 3	8 0 2	75 K= 15	57 7 0 L 50 9	20 -720 =6 62 62 63 -963	-13 0 0	4 6 8 10 12	741 234 1082 369 +137	683 -6 254 -2 1079 10 318 - 110 -	683 254 079 318 110	00000
8 10 12 14	1991 2095-2095 760 806 -806 660 629 629 297 317 317 616 579 579		K 2 +1 4 8 6 2 8 +1	-4 L·-2 18 53 51 02 873 871 57 256 -206 - 28 82 71 6 82 71	-16 58 -151 41	7 9 0	649 721 K-4 475	687 -685 824 820 L-4 524 520	-53 82 -62	4 6 8	+11 108 24 K=	8 611 7 2	99 -99 18 1118 89 -289	0	1 3 5	K=1 +131 +128 968	L= -8 12 301 - 920 -9	-5	-11 22 -28
1 3 5	K-1 L1 75 65 63 1005 1230 1230 469 388 387	-15 -28 -14	K 1 7 3 5	·5 L2 01 719 -700 - 22 591 584 B1 286 269	-163 88 -97	46	149 848 K=0 267	91 -7 919 -913 L= -4	91 -111	3 5 7 9	95 83 44	1 9 5 8 18 3	02 -902 12 812 97 397 54 -454	-16 0 -21 -11	9 11	784 597 605 K-2	764 7 554 5 528 - 5 L= -8	554 528	-16 -8
7 9 11 13 15	1952 2167-2167 297 203 202 375 401 399 697 668 668 +129 33 8	-35 14 -29 12 32	K 0 21 2 17 4 10	-0 L-3 36 1971-1971 17 1636 1636 00 1020 1020	0 0 0	4 6 10 12	2111 868 544 595 126	2113-2113 739 739 543 543 581 581 49 49	0000	0 2 4 6	K- +11 92 +12 95	2 L 7 2 8 7 3 2 10	-6 51 41 57 -857 92 -87 28 1028	30 -1 -31 16	4 6 8 10	688 261 1021 329	623 -6 232 -2 995 9 299 -1	523 230 295	-30 -6 -46
2 4 6 0	K=2 L+ -1 309 427 425 1118 1299 1299 1345 1621-1619	-46 10 -19 1; -75 1	68 82 03 26 4	37 796 -796 68 286 -286 34 318 -318 72 716 716 13 417 417	0 0 0 0	14 16 1	1099 191 K=1 1127	1076-1076 173 173 L= -4 1159 1159	0	1 3 5	K≈ +13 77 69	3 L 0 1 1 7 9 6	-6 23 -106 48 -747 95 695	64 -43 1	2 * 6 8	K=0 803 843 337 +139	L= -9 681 6 781 -7 244 -2 102 -1	581 781 244 102	0000
10 12 14	567 557 556 284 283 274 559 540 533	3 15 -70 82	к 15 315 53	-1 L-3 49 448 -447 93 1803 1803 27 268 -268	-29 6 - 17	357911	1075 1036 996 412 770 938	1129-1129 1072-1072 941 941 460 460 672 670 889 -890	-6 16 23 -16 46	24	K= 14 126 26	0 L 4 1 5 1	-6 98 -98 50 1350 88 -188	0	3 5 7	K-1 +140 786 +140	L= -9 55 - 760 -7 85 -	-55 759 -83	-8 -33 18
						- ,	-)0	-009	3	0		1 1	-101	v					

O-O distance for adjacent oxygens on unshared edges varies from 2.73-3.26 Å. Although the average bond angle involving two adjacent oxygens and either metal is very close to the undistorted octahedral value of 90°, the range is from 72.7° to 108.6° .

It is interesting to note that for both gallium and niobium the four shortest metal-oxygen bonds (first four distances in Table 3) form a distorted tetrahedron. For both metals the average bond angle (first six bond angles in Table 3) for this distorted tetrahedral group



Fig. 2. Upper part of the figure illustrates a unit of the hypothetical R_3O_8 structure formed by fusing together the blocks of octahedra. Each hatched square represents an octahedron in projection as viewed down a corner; the lighter lines indicate blocks diplaced downwards by half an octahedral diagonal. Two of these R_3O_8 units can be joined together (by translating the lower unit sideways by $1\frac{1}{2}$ octahedral diagonals) to form the double octahedral sheet.

is within a few degrees of the tetrahedral angle, but one of the six angles in each case is very large (about 149°). In Fig. 1, the longer metal-oxygen bonds are shown by broken lines. Tetrahedrally coordinated niobium is reported by Komkov (1959) in fergusonite (natural YNbO₄). The crystals display an excellent cleavage parallel to (001) and a good cleavage parallel to (100). Both cleavages involve rupturing of the longer, and thus weaker, metal-oxygen bonds.

Another interesting feature of the structure is the rather large void centered at x=0, $y=\frac{1}{2}$, $z=\frac{1}{2}$. The shortest distance between this point and a neighboring oxygen position is 2.62 Å. An ion with a radius somewhat greater than 1 Å could be easily accommodated in this void. No structures analagous to that of GaNbO₄ with the void occupied are known to the authors.

Wadsley (1961b) proposes a structural scheme for a series of disordered titanium niobium oxides of composition $R_{3n}O_{8n-3}$. This series includes both known and hypothetical members. The structures are formed from units consisting of three octahedra forming a short chain by sharing of corners. These units are linked into 'single' or 'double' layers in the manner shown in Fig. 2 by edge and corner sharing. The several mem-

Table	3.	Interatomic	separations	and	angles
	~ •			~~~~~	W1112 100

$\begin{array}{c} Ga - O(4a) \\ Ga - O(3a) \\ Ga - O(1b) \\ Ga - O(1c) \\ Ga - O(1d) \\ Ga - O(2a) \end{array}$	1.891 Å	Nb-O(4a)	1·797 Å
	1.918	Nb-O(3e)	1·906
	1.922	Nb-O(2c)	1·937
	2.021	Nb-O(2b)	2·000
	2.097	Nb-O(1a)	2·215
	2.125	Nb-O(3f)	2·333
O(2a)-O(1c)	2·498	O(3e)-O(3f)	2·631
O(1a)-O(2c)	2·498	O(1d)-O(1b)	2·666
O(2a)-O(1b)	2·583	O(1d)-O(1c)	2·666
O(1a)-O(2b)	2·583	O(3a)-O(1d)	2·698
$\begin{array}{c} O(4a)-O(2c)\\ O(3a)-O(4a)\\ O(3f)-O(2b)\\ O(4a)-O(1c)\\ O(4a)-O(3e)\\ O(2a)-O(4a)\\ O(2a)-O(4a)\\ O(4a)-O(2b)\\ O(3f)-O(2c) \end{array}$	2·730 2·800 2·840 2·879 2·894 2·895 2·895 2·896 2·924	$\begin{array}{c} O(3e)-O(2c)\\ O(3a)-O(1c)\\ O(4a)-O(1b)\\ O(2a)-O(1d)\\ O(3e)-O(2b)\\ O(3a)-O(1c)\\ O(1a)-O(4a) \end{array}$	2·934 2·959 2·965 2·969 3·096 3·109 3·264
$\begin{array}{l} O(3a)-Ga-O(4a)\\ O(4a)-Ga-O(1c)\\ O(3a)-Ga-O(1b)\\ O(4a)-Ga-O(1b)\\ O(3a)-Ga-O(1c)\\ O(1c)-Ga-O(1b) \end{array}$	94·51°	O(4a)-Nb-O(2c)	93.89°
	94·71	O(4a)-Nb-O(2b)	99.28
	100·78	O(3e)-Nb-O(2b)	99.55
	102·22	O(4a)-Nb-O(3e)	102.78
	104·06	O(3e)-Nb-O(2c)	104.85
	148·60	O(2b)-Nb-O(2c)	148.80
$\begin{array}{l} O(2a)-Ga-O(1b)\\ O(2a)-Ga-O(1c)\\ O(1d)-Ga-O(1c)\\ O(1d)-Ga-O(1b)\\ O(3a)-Ga-O(1d)\\ O(2a)-Ga-O(1d)\\ O(2a)-Ga-O(1d)\\ O(2a)-Ga-O(4a) \end{array}$	76-12 77-03 80-66 83-06 84-21 89-38 92-07	$\begin{array}{c} O(1a)-Nb-O(3f)\\ O(1a)-Nb-O(2b)\\ O(1a)-Nb-O(2c)\\ O(3e)-Nb-O(3f)\\ O(3f)-Nb-O(2c)\\ O(3f)-Nb-O(2c)\\ O(3f)-Nb-O(2b)\\ O(4a)-Nb-O(1a) \end{array}$	72.73 73.64 75.39 75.98 81.53 85.90 108.46

In parenthesis for each oxygen atom are letters which refer to the following symmetry operations:

$$\begin{array}{rcl} a & x, y, z \\ b & \frac{1}{2} - x, \frac{1}{2} + y, 1 - z \\ c & \frac{1}{2} - x, -\frac{1}{2} + y, 1 - z \\ d & x, y, 1 + z \\ e & -x, -y, 1 - z \\ f & x, y, -1 + z \end{array}$$

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bers of the series may now be formed by cornersharing between successive single and double layers in the appropriate ratio. A structure consisting of purely single layers has the composition R_3O_8 (not a part of the series).

The composition, $GaNbO_4$ or R_2O_4 , does not appear in Wadsley's proposed series. The configuration of shared octahedra in GaNbO₄, however, bears a strong resemblance to part of Wadsley's double layers. A structural scheme is here proposed which will include a compound of the R_2O_4 type with a structure corresponding to that of GaNbO₄. This scheme differs from Wadsley's in that the fundamental unit consists of two rather than three octahedra. The series of composition $R_{4n}O_{10n-4}$ is formed, then, by assembling the units into single and double layers and linking these in appropriate sequence as shown in Fig. 3. A structure consisting of single layers alone would have the composition R_2O_5 , again not fitting into the series. If n=2, we have R_8O_{16} consisting only of double layers. If we consider this to be an ordered structure involving two metals we can write Ga₄Nb₄O₁₆, the unit-cell contents for the present compound. Other members of the series would be:

$$\begin{array}{l} n = 3 \\ n = 4 \end{array} \quad \begin{array}{l} R_{12}O_{26} \\ R_{16}O_{36} \\ n = tc. \end{array}$$

The compound R_4O_6 , for n=1, cannot be made by any combination of double and single layers. No structures of this type corresponding to these compounds are known to us.

Wadsley proposes both a monoclinic and orthorhombic structure in his series with n=6. This compound has the composition R_2O_5 . In the scheme proposed here the single-layer structure has the composition R_2O_5 and would be orthorhombic, *C*-centered, with dimensions near $a_0 = 12.4$, $b_0 = 3.7$, $c_0 = 4.2$ Å. No detailed structures for Nb₂O₅ or Ta₂O₅ have been reported, but several reported cell dimensions are related to these values or to simple multiples or fractions thereof (Lapitskii, Simanov, Semenenko & Yarembash, 1954; Simanov, Lapitskii & Artamonova, 1954).

The assistance of Mr H. Bezdek in collecting the intensity data, the useful discussions with Dr J. R. Brathovde (National Science Foundation), the computer assistance of Dr F. H. Kruse (Los Alamos



Fig. 3. The hypothetical R_2O_4 structure is formed with two octahedra compared with three for R_3O_8 (Fig. 2). Two of these R_2O_4 units (top of figure) can be joined together by translation of the lower unit sideways by $\frac{1}{2}$ octahedral diagonals to form the double layer R_8O_{16} (lower figure) of which GaNbO₄ is a member.

Scientific Laboratory), and the piezoelectric examination by Dr Robert Douglas (Los Alamos Scientific Laboratory), are acknowledged.

APPENDIX

The referee has objected to our choice of the noncentric space group, especially since the difference of the y coordinates from zero between a centric and noncentric space group would probably be strongly correlated with thermal displacements parallel to b_0 (*i.e.*, the B_{22} term using anisotropic thermal parameters). We have refined, with several cycles of differential synthesis, the positional and anisotropic thermal parameters for a centric model. The final R valueof 0.078 using only observed reflections (0.092 for all reflections) was not significantly different from that obtained with the noncentric case. These centric parameters are presented in Table 4*.

Table 4.	Atomic p	positions	and ani	sotropic
temperatur	e factors	for cent	tric inter	pretation

	x	Z	B_{11}	B_{22}	B_{33}	B_{13}
Ga	0.19712	0.81359	0.084	0.303	0.044	0.082
Nb	0.10320	0.23062	0.302	0.449	0.237	0.058
O (1)	0.25941	0.14594	0.312	0.331	0.294	-0.012
O(2)	0.36252	0.79879	0.487	0.082	0.295	-0.047
O(3)	0.05492	0.86129	0.164	0.799	0.325	0.190
O(4)	0.13705	0.51397	0.567	1.319	0.104	0.011

The editor has pointed out that a y coordinate of 0.025, representing a displacement of the oxygen atoms from the symmetry plane by about 0.1 Å, is essentially indistinguishable from a y coordinate of zero and a B_{22} term of such a magnitude as to give an extra 0.1 Å r.m.s. thermal displacement along b. This appears to be the case for oxygen atoms O(3) and $\overline{O(4)}$. The average of the difference between the r.m.s. displacements along b_0 and the next largest principal axis for these atoms is 0.06 Å. This perhaps would be predictable since these oxygen atoms do not have any constraining bonds directed along b_0 whereas O(1) and O(2) do. In addition the anisotropic thermal parameters for O(1) and O(2) are in general agreement with their respective regular and irregular arrangement of nearest neighbors. Similarly the differences of the r.m.s. displacement along b_0 and the next largest principal axis for the Ga atom corresponds very closely to the 0.03 Å displacement from the symmetry plane obtained in the noncentric refinement.

A satisfactory solution for our intensity data may be obtained either with the centric space group with anisotropic refinement of thermal parameters or with the noncentric space group and isotropic thermal para-

^{*} A table of observed and calculated structure factors has not been included for the centric refinement. Values are not significantly different from those listed in Table 2. Interatomic separations are essentially the same as Table 3 with approximate averages to be taken between atoms related by the symmetry operations (b) and (c).

meters. We believe that where there are two solutions possible, it is preferable to choose the more general case, *i.e.* the noncentric interpretation.

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The Crystal Structure of NaCl. $5\frac{1}{7}$ NH₃ and NaBr. $5\frac{1}{7}$ NH₃

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The crystal structures of NaCl. $5\frac{1}{7}$ NH₃ and NaBr. $5\frac{1}{7}$ NH₃ have been determined from single-crystal X-ray data, collected in a low-temperature Weissenberg camera with Cu K radiation. Both compounds are trigonal, space group $R\overline{3}$, with 21 formula units in a hexagonal cell for which the measured dimensions are:

NaCl. $5\frac{1}{7}$ NH₃: a = 23.67, c = 9.25 Å (t = -58 °C); NaBr. $5\frac{1}{7}$ NH₃: a = 24.34, c = 9.46 Å (t = -35 °C).

The ammonia molecules and halide ions together form a roughly cubic close-packed arrangement, the sodium ions occupying certain of the octahedral interstices, so that only ammonia molecules are closest neighbours. The octahedra, which enclose some of the sodium ions, share two corners with adjacent octahedra; in this case the effective coordination of the sodium is only fivefold, the sixth ammonia molecule lying at a greater distance.

Introduction

The present investigation forms part of a series of structure determinations of some ammines formed with salts of the alkali metals and ammonium. The structures of $NH_4 . 3NH_3$ (X=Cl, Br, I) and $NH_4I . 4NH_3$ have been reported earlier (Olovsson, 1960 *a*, *b*)

In the systems NaX–NH₃ (X=F, Cl, Br, I) the following compounds have been reported: NaCl . $2 \cdot 5$ NH₃ (Friedrichs, 1921), NaCl . 5NH₃ (Joannis, 1891; Biltz & Hansen, 1923; Biltz, 1923; Patscheke, 1933), NaBr . $5 \cdot 25$ NH₃ and NaBr . $5 \cdot 75$ NH₃ (Biltz & Hansen, 1923; Biltz, 1923), NaI . 4NH₃ (Picon, 1919; Leonard, Lippincott, Nelson & Sellers, 1955), NaI . $4 \cdot 5$ NH₃ and NaI . 6NH₃ (Biltz & Hansen, 1923; Biltz, 1923). A phase with the composition NaCl . $2 \cdot 5$ NH₃ does not exist according to later investigations (Biltz & Hansen, 1923). The phase reported as NaI . 4NH₃ has the composition NaI . $4 \cdot 5$ NH₃ according to Biltz & Hansen (1923). The structures of the ammines of sodium iodide are currently being investigated in this laboratory.

In this paper the structures of the compounds formulated above as NaCl. $5NH_3$ and NaBr. $5\cdot25NH_3$ are reported. The X-ray crystallographic investigation shows that these compounds are isomorphous and have the exact composition NaCl. $5\frac{1}{7}NH_3$ and NaBr. $5\frac{1}{7}NH_3$, respectively. This point is further discussed below. For covenience these compounds are called 'pentammines' in the following text, although it should be kept in mind that this designation is not altogether adequate.

Experimental

The solubility curve of sodium chloride in ammonia is shown in Fig. 1. The values were taken from the work by Patscheke (1933) (*cf.* also Patscheke & Tanne, 1935), but the results of Guyer, Bieler & Schmid (1934) and Distanov (1938) are very similar. The curves give the composition of the solution in equili-